

Hydrolytic Degradation of Thermoplastic Copolyester Nanocomposites



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Abstract

Thermoplastic copolyesters (TPE-Es) are multi-block copolymers commonly incorporating hard (polyester) and soft (polyether) blocks. Multifunctional epoxy-based chain extenders (TGDDM) are one of the additives that serve to reverse molecular weight (MW) damage caused by the process of hydrolysis of TPE-E. There are two functional groups that are capable of additional reactions on the terminal hydroxyl and carboxyl groups of TPE-E. Considering the hydrogen bonded hard segments in hard domains are less susceptible to hydrolysis compared to the non-hydrogen bonded equivalents, the phase separated morphology of polyether-polyester copolymers based TGDDM also provides to the long-term stability. The molecular weight of TPE-E was determined by GPC and the thermal and tensile behaviors were investigated.

Keywords: Thermoplastic copolyester; Crosslinking; Hydrolysis; Mechanical properties

Abbreviations: TPE-Es: Thermoplastic Copolyesters; MW: Molecular Weight; TGDDM: Tetraglycidyl-4,4-Diamino-Diphenylmethane;

Introduction

Polyester hydrolysis produces hydrophilic acid (carboxyl) end groups in the polymer, which increase the ability for the polymer to absorb water because of the decrease in its surface resistance and hydrophobicity [1-3].

Chain extenders remain necessary in improving the molecular weight of TPE-E and, consequently, broaden their industrial applications. Chain extenders help in the substitution of polyester hydrolysis-caused molecular weight damage, leading to the production of a polymer with desired melting strength and rheological properties. The chain extenders function effectively, predominantly due to their bi-functional groups that inherently react with polymer terminal groups. Moreover, specific chain extenders,

such as tetraglycidyl-4,4-diamino-diphenylmethane (TGDDM), are vital in the enhancement of TPE-E molecular weights. Consequently, TPE-E would have improved rheological, mechanical, and thermal characteristics [4].

Hytrel® 3078 is a brand of TPE-E (copolyesters, COPEs) under the trademark of DuPont. Hytrel 3078 nanocomposite prepared by reactive extrusion-based masterbatch (HC1-C2), dry feed (HC2) of organo-fluoromica combined with TGDDM, the post-addition of TGDDM (HC1-T) and the direct-addition of TGDDM-organofluoromica (HT-C1) were compared for their resistance to hydrolysis in aqueous buffer solutions for 140 h solution at 100°C at pH 1 (Figure 1).

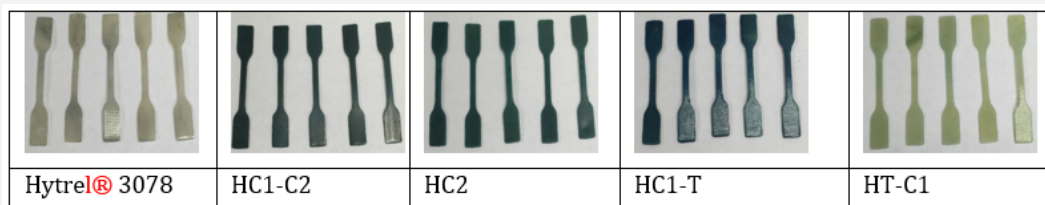


Figure 1: The hydrolysis experiments carried out by immersing dumbbell samples (5 replicates) in pH 1 aqueous buffer solutions for 140h at 100°C.

The enlargement of the MWD for the nanocomposites can also be indicated by the polydispersity index ($PDI = M_w/M_n$), besides extremely high molecular weights above 2×10^6 g/mol are also obtained. These outcomes show modifier chain extension

efficiency which has an effect on the rheological characteristics of the material.

The PDI of the host polymer before and after reactive modification via different processing routes (blank TPE-E, HT-C1,

HC2, HC1-C2 and HC1-T) are also given in Table 1. A significant decrease in M_n of HC1-C2 nanocomposites occurred during a masterbatch of reactive nanofiller in TPE-E. It is still expected that there may be thermal and thermo-oxidative degradation processes associated with reactive processing this TPE-E, which are responsible for the molecular weight decrease in the melt compounded TPE-E. The relatively low M_w , suggesting that HC1-C2 is less crystalline than other modified nanocomposites, and therefore it was not possible to obtain high molecular weight TPE-E. Higher molecular weight may be obtained if the TGDDM is post-added with TPE-E matrix for HC1-T. It was shown that the M_n of the HC1-T was twice that of first adding of TGDDM into TPE-E (HT-C1). This halving of molecular weight resulted in the HT-C1 having a tensile strength of 35MPa, which was 4.9MPa lower than that of the HC1-T (39.9MPa) (Figure 2). The post-addition of TGDDM decreased the PDI of TPE-E nanocomposite.

This contrasts with the significant improvement in M_w of HC1-C2. The high PDI value can also be attributed to higher branching formation of TPE-E. Twin screw extrusion, which compounds thermally-sensitive materials, yields a far greater shear energy than single screw extrusion, which results in a loss of molecular weight and a decrease in properties and performance [5].

Table 1: Characteristic molecular weights obtained by GPC of TPE-E nanocomposites.

Materials	M_n	M_w	M_z	Polydispersity (PDI)
Blank	46,056	73,177	112,697	1.58
HT-C1	57,382	101,986	172,302	1.77
HC2	56,713	106,106	183,313	1.87
HC1-C2	38,947	72,522	127,113	1.86
HC1-T	94,225	140,520	207,950	1.49

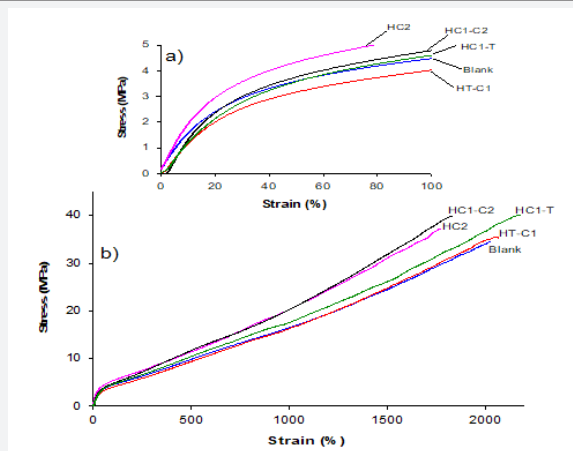


Figure 2: Effect of modification route on the (a) modulus, and (b) stress-strain curve of TPE-E nanocomposites.

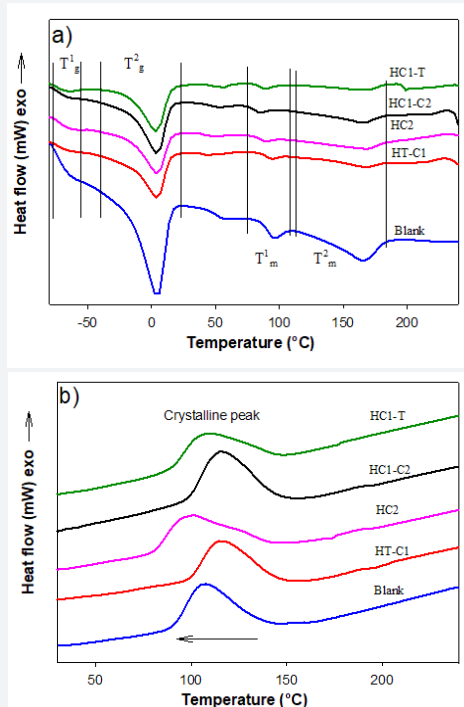


Figure 3: DSC thermograms of (a) heating scan and (b) cooling scan of TPE-E nanocomposites.

The thermal and mechanical properties of polymers are influenced by polydispersity of the hard segments. The presence of many epoxy groups gives rise to a highly cross-linked network within the nanocomposite. As seen in Figure 3, the addition of TGDDM in HT-C1, HC1-C2 and HC1-T clearly shifted the exothermic heat of crystallisation peaks of the hard segment to higher temperatures. The result of these studies indicates that hard segments of uniform length crystallise faster and better than hard segments of non-uniform length. However, the hard segments represent a physical cross-link which report elastomeric properties to the soft phase. The hard-soft segmental solubility parameters and aligned hard and soft segments, polydispersity and molecular weight, as well as the processing and thermal history of the TPE-E host matrix are essential factor to be considered to the tensile properties demonstrated by the nanocomposites [6].

References

1. Choi T, Weksler J, Padsalgikar A, Hernández R, Runt J (2009) Polydimethylsiloxane-Based Polyurethanes: Phase-Separated Morphology and *In Vitro* Oxidative Biostability. *Aust J chem* 62(8): 794-798.
2. Kelleher G, Bebbington H, Falcone R, Ryan T, Went P (1979) *Soc Plast Eng Tech Papers* 25: 527.
3. Amin M, Akbar M, Amin S (2007) Hydrophobicity of Silicone Rubber used for Outdoor Insulation (An Overview). *Rev Adv Mater Sci* 16: 10-26.
4. Roh JH, Kim D, Lee HD, Yoon BK (2012) Preparation of high molecular weight and elastic copolyester by reactive extrusion with diisocyanate compound for laser printing process. *J Polym. Eng* 32(3): 207-214.
5. Dennis HR, Hunter DL, Chang D, Kim S, White JL, et al. (2001) Effect of melt processing conditions on the extent of exfoliation in organoclay-based nanocomposites. *Polymer* 42(23): 9513-9522.
6. Miller JA, Lin SB, Hwang KK, Wu KS, Gibson PE, et al. (1985) Properties of polyether-polyurethane block copolymers: effects of hard segment length distribution. *Macromolecules* 18(1): 32-44.



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